

Video Article

Fabrication of 109° Periodic Domain Walls with a Bottom Electrode in BiFeO₃ Thin Films

Deyang Chen^{*1}, Chao Chen^{*1}, Peilian Li¹, Zhen Fan¹, Xingsen Gao¹, Jun-Ming Liu^{1,2}

¹Institute for Advanced Materials, South China Academy of Advanced Optoelectronics, South China Normal University

²Laboratory of Solid State Microstructures and Innovation Center of Advanced Microstructures, Nanjing University

*These authors contributed equally

Correspondence to: Deyang Chen at dychen1987@gmail.com, Xingsen Gao at xingsengao@scnu.edu.cn

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Abstract

A variety of exciting phenomena have been discovered using 109° domain walls in BiFeO₃ thin films, such as domain wall conductivity, photovoltaic effects, and magnetoelectric coupling effects. The control of these physical properties with an electric field plays a key role in the development of nanoelectric devices. Therefore, it is critical to fabricate 109° periodic domain walls with a bottom electrode. However, the introduction of a bottom electrode favors the formation of a 71° domain structure due to the electrostatic boundary conditions. In this study, pulsed laser deposition (PLD) is used to produce multilayer epitaxial thin films. A 25% La doping BiFeO₃ layer is inserted as the dielectric spacer between the bottom electrode SrRuO₃ layer and the pure BiFeO₃ layer, enabling the fabrication of 109° periodic domain walls engineered by an interface effect-depolarization field. Moreover, the fabrication of the 109° domain structure with a bottom electrode enables the study of its switching behavior. This protocol provides a novel route to produce 109° periodic domain walls and opens a new pathway to explore fascinating phenomena, such as the room-temperature electric field control of exchange bias in a ferromagnet/BiFeO₃ system and room-temperature multiferroic vortices in BiFeO₃.

Introduction

Domain wall functionalities in BiFeO₃ thin films, such as domain wall conductivity¹, photovoltaic effects², magnetism³, and magnetoelectric coupling⁴, have inspired many studies on the fabrication and manipulation of the domain structures^{5,6,7}. Periodically ordered 71°, 109°, and 180° stripe domains have been obtained by tuning the film thickness effects, misfit strain effects, and electrostatic boundary conditions^{8,9,10,11}. The periodic 71° stripe domain structure can be produced on thick SrRuO₃ bottom electrodes and can be reversibly controlled under an electric field⁸, which has promoted a series of seminal works on the electric-field control of magnetism in the ferromagnet/multiferroic BiFeO₃ system^{4,12,13,14,15}. However, the pure 109° stripe domain structure can only exist without a (or with an ultra-thin) SrRuO₃ bottom electrode⁸, making it typically unstable or unresponsive under an applied electric field and inhibiting the study of the switching behavior of 109° domains. Moreover, a large exchange bias has been demonstrated in the ferromagnet/BiFeO₃ system with the 109° BiFeO₃ domain walls^{16,17}. Thus, the successful fabrication of a 109° domain structure with a bottom electrode is promising for controlling exchange bias using an electric field. This has great potential application to low-energy-consumption, non-volatile magnetoelectronic memory devices.

Our previous study¹⁸ revealed a novel approach to precisely control the domain structure in BiFeO₃ thin films by tuning the depolarization field with a dielectric layer. In the BiFeO₃/SrRuO₃/DyScO₃ stack without La-BiFeO₃, the screening effects at the ferroelectric (BiFeO₃)/metallic (SrRuO₃) interface enable the formation of 71° domains, while the introduction of the La-BiFeO₃ dielectric space layer enables an increase in the distance between the screening charges from the SrRuO₃ and BiFeO₃. This leads to a reduction of the screening effects and thus an increase in the depolarization field. Consequently, the 71° stripe domain is destabilized and a 109° stripe domain structure forms to decrease the energy cost due to the strong depolarization field.

In this work, we focus on the fabrication of 109° periodic domain walls with a bottom electrode in BiFeO₃ thin films by introducing a La-BiFeO₃ dielectric layer. The detailed protocol describes how to grow the multilayer thin films using PLD and how to use piezoresponse force microscopy (PFM) to measure the 109° domain structure and study its switching behavior. The typical PLD system is shown in **Figure 1**. This protocol can help to increase the reproducibility of the fabrication of 109° periodic domain walls and promote the study of domain wall functionality.

Protocol

1. Substrate Preparation

1. Clean a 5 mm × 5 mm × 0.5 mm single-crystal DyScO₃ (110) substrate with acetone in an ultrasonic cleaner for 5 min.

2. Rinse the substrate with acetone for 5 s and transfer it to isopropyl alcohol. Clean it for 5 min in the ultrasonic cleaner.
 3. Rinse the substrate with isopropyl alcohol for 5 s and dry it using N_2 flow.
 4. Mount the substrate on a heater with silver paint and then put the heater on a hot plate to dry the silver paint for 10 min at 100 °C.
 5. After the heater cools down to room temperature, blow it with N_2 flow. Mount it into the PLD chamber.
- NOTE: The protocol can be paused here.

2. PLD Setup

1. Mount a $SrRuO_3$ target, a $BiFeO_3$ target, and a 25% La doping $BiFeO_3$ (La- $BiFeO_3$) target in the PLD chamber.
2. Set the target-to-substrate distance to 50 mm.
3. Align the laser light path to focus the laser spot on the target. Attach a piece of sensitive paper on the target and move the focusing lens to a proper position to obtain uniform laser spots, which will ensure the homogeneity of the laser energy.
NOTE: The target will rotate with roto-translational motion to avoid overheating the target during film growth.
4. Measure the laser energy outside and inside the laser window using a power meter to calculate the energy loss rate.
NOTE: The loss rate increases with increasing growth time due to the coating on the laser window during deposition. Here, the laser window is cleaned every other day.
5. Pump the chamber with a mechanical pump until the vacuum is less than 10 Pa. Use the turbo pump to obtain a high vacuum ($< 5 \times 10^{-4}$ Pa).
NOTE: The protocol can be paused here.

3. Selection of Deposition Parameters

1. Adjust the pumping speed by tuning the gate valve. Inflate oxygen into the chamber to create oxygen pressure of 13 Pa.
2. Heat up the heater to 700 °C at a rate of 20 °C/min and anneal the substrate for 10 min.
3. Start the pulsed laser and set the desired laser energy for $SrRuO_3$ growth by changing the voltage. Measure the laser energy using a power meter at a laser frequency of 2 Hz.

4. Growth of Multilayer Films

1. Clean the $SrRuO_3$ target surface at a frequency of 10 Hz for 3 min.
2. Grow $SrRuO_3$ epitaxial thin film on a (110)-oriented $DyScO_3$ substrate at 10 Hz for 6 min using PLD.
NOTE: The growth temperature should be 700 °C and the oxygen partial pressure should be 13 Pa.
3. Tune the heater temperature to 690 °C and clean the La- $BiFeO_3$ target surface at a frequency of 10 Hz for 3 min.
4. Start the growth of the La- $BiFeO_3$ layer for 10 min at 10 Hz.
NOTE: The growth temperature should be 690 °C and the oxygen partial pressure should be 13 Pa.
5. Clean the $BiFeO_3$ target surface at a frequency of 10 Hz for 3 min.
6. Adjust the frequency of the pulsed laser to 5 Hz and start to grow the $BiFeO_3$ layer for 40 min at 690 °C under oxygen partial pressure of 13 Pa.
7. Close the gate valve and inflate oxygen into the chamber. Cool down the sample at 10 °C/min in a 10,000-Pa oxygen atmosphere.
8. Open the chamber after the heater temperature is below 80 °C and remove the sample from the heater.

5. PFM Measurement

1. Attach the sample to a thin metal plate and contact the bottom electrode with the plate using silver paint.
2. Take the PFM measurement of the sample with the cantilever along (110) under the PFM mode to capture 109° domain structure images.
NOTE: The PFM scanning size should be $5 \mu m \times 5 \mu m$.
3. Apply -6 V in a $3 \mu m \times 3 \mu m$ box to study the switching behavior of the 109° domain structure.
4. Capture the $5 \mu m \times 5 \mu m$ PFM images after -6-V switching.

Representative Results

The multilayer $BiFeO_3$ /La- $BiFeO_3$ / $SrRuO_3$ films on the $DyScO_3$ (110) substrate are produced by PLD, and the heterostructure stack is shown in **Figure 2**. To obtain the 109° domain structure with a bottom electrode, a thin dielectric layer of La- $BiFeO_3$ is inserted between the $SrRuO_3$ bottom electrode and the ferroelectric $BiFeO_3$ layer.

As demonstrated in our previous work¹⁸, the thickness of the La- $BiFeO_3$ layer is crucial to the formation of pure 109° domain walls. An ultra-thin La- $BiFeO_3$ layer (< 10 nm) would lead to mixed 71° and 109° domain walls. In this protocol, thicker La- $BiFeO_3$ film layers (> 10 nm) are grown to achieve a pure 109° domain structure.

By carefully controlling the PLD growth parameters, including the laser path alignment, laser energy, oxygen pressure, and heater temperature, 109° domain walls can be fabricated in the $BiFeO_3$ /La- $BiFeO_3$ / $SrRuO_3$ / $DyScO_3$ sample, as shown in the PFM data (**Figure 3**). The switching behavior of the 109° domain structure has also been studied by applying a -6-V voltage to the sample, as displayed in **Figure 4**. It is revealed that the 109° domain structure with a bottom electrode can be switched to a 71° domain structure.

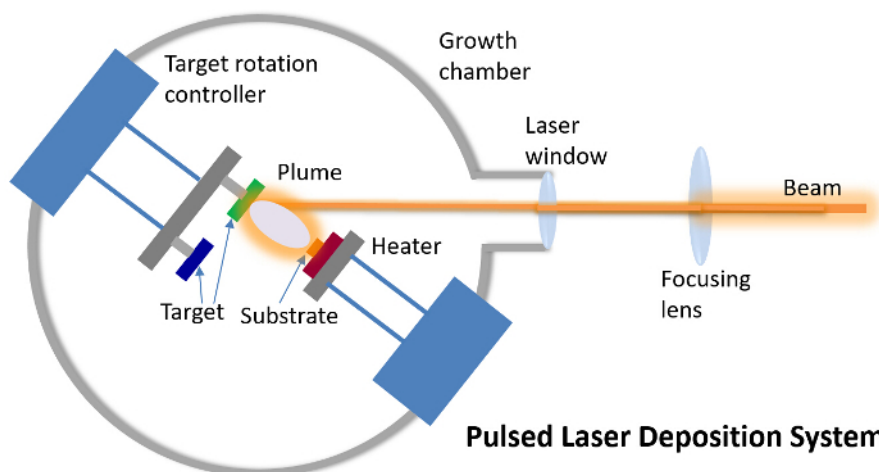


Figure 1. Schematic of the PLD system. [Please click here to view a larger version of this figure.](#)

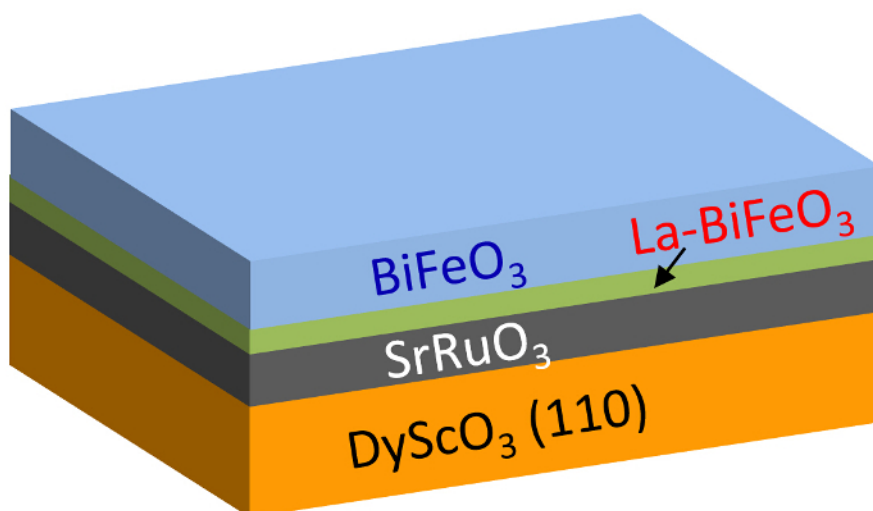


Figure 2. Schematic of the heterostructure $\text{BiFeO}_3/\text{La-BiFeO}_3/\text{SrRuO}_3/\text{DyScO}_3$ stack. [Please click here to view a larger version of this figure.](#)

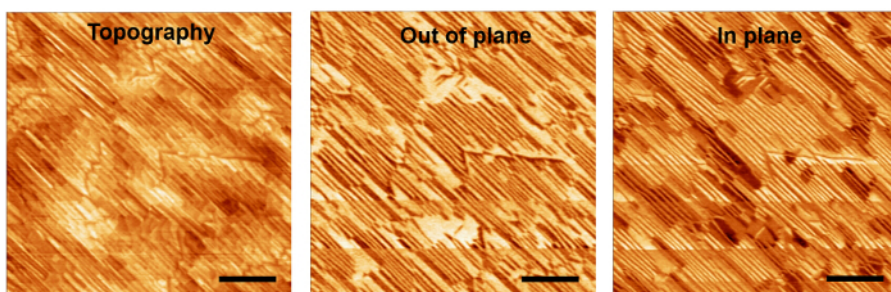


Figure 3. Topography and out-of-plane and in-plane PFM images of the obtained sample with a 109° domain structure. The size of the images is $5\ \mu\text{m} \times 5\ \mu\text{m}$. Scale bar: $1\ \mu\text{m}$. [Please click here to view a larger version of this figure.](#)

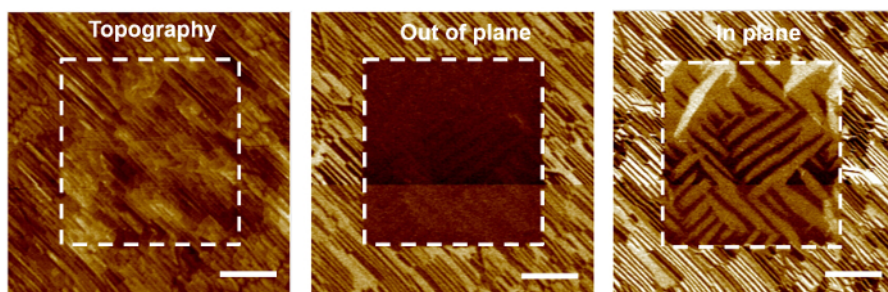


Figure 4. Switching behavior of a 109° domain structure.

Topography and out-of-plane and in-plane PFM images after a -6-V voltage switch. The size of the images is 5 μm × 5 μm, and the size of the switched regions within the dashed squares is 3 μm × 3 μm. Scale bar: 1 μm. [Please click here to view a larger version of this figure.](#)

Discussion

PLD is a powerful technique to fabricate complex oxide epitaxial thin films¹⁹. Using this technique, many investigations have been carried out on BiFeO₃ thin films^{13,20,21,22}. As one of the most striking aspects, domain walls are widely studied due to a wealth of fascinating phenomena^{1,2,3}, such as domain wall conductivity and enhanced magnetism at a 109° domain wall. However, a 109° domain structure can only exist without a (or with an ultra-thin) bottom electrode⁸, inhibiting the study of electric field control of the related physical phenomena. In this study, 109° periodic domain walls with a bottom electrode are produced in BiFeO₃ thin films by PLD, which indicates the significance with respect to existing methods⁸.

The PLD growth conditions (including laser energy, growth temperature, and oxygen pressure) of high-quality BiFeO₃, La-BiFeO₃, and SrRuO₃ thin films are critical to this protocol. In addition, the selection of the La doping content plays a key role in obtaining the 109° domain structure. Our unpublished data show that La-BiFeO₃ is still ferroelectric when the La doping content is less than 18%, which would not lead to the formation of 109° periodic domain walls in the BiFeO₃/La-BiFeO₃/SrRuO₃/DyScO₃ stack. Non-ferroelectric 25% La-BiFeO₃ is used as a dielectric layer in this study.

One of the limitations of this technique is that the maximum size of the sample can only be 1 cm × 1 cm; otherwise, the film would not be uniform. The substrate selection of DyScO₃—which offers anisotropic strain, excluding two of the possible structural variants, to yield a 109° stripe domain structure—is another possible limitation, while other substrates, including SrTiO₃, LaAlO₃, etc. cannot provide the required anisotropic strain.

Previous studies²³ demonstrated that, due to the interplay between strain, depolarization field, and gradient energies, topological ferroelectric vortices can be produced in PbTiO₃/SrTiO₃ superlattices. The method shown in this protocol provides a possible application to explore the room-temperature multiferroic vortices in the BiFeO₃ system. Furthermore, it can be used to control the ferroelectric polarization or domain structures in ferroelectric materials, such as BaTiO₃ and PbZr_xTi_{1-x}O₃.

Disclosures

The authors declare no conflicts of interest.

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